# Thermodynamic Relations among Olivine, Spinel, and Phenacite Structures in Silicates and Germanates. III. The System CuO-MgO-GeO<sub>2</sub>

A. NAVROTSKY

Department of Chemistry, Arizona State University, Tempe, Arizona 85281

Received October 3, 1973

Phase relations in the system CuO-MgO-GeO<sub>2</sub> in air at 1000°C were determined. Activitycomposition relations in the system CuO-MgO at 1000°C were measured: the guggenite phase was found to exist over the composition range  $0.695 < X_{CuO} < 0.805$ . The enthalpy and free energy of formation of CuGeO<sub>3</sub> were determined by high temperature solution calorimetry and reduction equilibria, respectively. For the reaction: CuO + GeO<sub>2</sub> = CuGeO<sub>3</sub>,  $\Delta G_{1213}^{\circ} = -4.0$  kcal/mole,  $\Delta H_{988}^{\circ} = -5.2$  kcal/mole,  $\Delta S^{\circ} = -0.9$  cal/deg mole. The stability of rocksalt, tenorite, olivine, enstatite, and CuGeO<sub>2</sub> structures in this system is discussed.

### 1. Introduction

As part of a continuing study of germanate systems (1, 2), this paper reports a determination of phase equilibrium and thermodynamic relationships in the system CuO-MgO-GeO<sub>2</sub> This system contains several features of interest. In the binary system CuO-MgO, a phase, exists, whose composition, guggenite,  $(CuMgO_2 (3), Cu_2MgO_3 (4), or Cu_3MgO_4 (5))$ and structure still remain open to question. The corresponding phase in the system CuO-CoO has been found to have a significant homogeneity range at 1000°C (0.63  $< X_{CuO} <$ 0.68) and activity-composition relations have been studied in that system (6). In the binary system CuO-GeO<sub>2</sub>, CuGeO<sub>3</sub> is the only compound formed (7). This is a "turquoise blue" material, whose structure (8) contains chains of almost square planar CuO<sub>4</sub> groups. The nonexistence of the orthogermanate  $Cu_2GeO_4$ , or of any compounds in the system  $CuO-SiO_2$  (9), can be attributed to the energetically unfavorable octahedral coordination of Cu<sup>2+</sup> in the structures (olivine, spinel, or pyroxene) normally encountered at these compositions in other systems. A study of the ternary system CuO-MgO-GeO<sub>2</sub>, by determining the extent of copper substitution into the olivine,  $Mg_2GeO_4$  and the pyroxene,

Copyright © 1974 by Academic Press, Inc. All rights of reproduction in any form reserved. Printed in Great Britain MgGeO<sub>3</sub> and by making thermodynamic calculations for these solid solutions, would provide a quantitative measure of the instability of CuGeO<sub>3</sub> (pyroxene) and Cu<sub>2</sub>GeO<sub>4</sub> (olivine).

In the present work, we have carried out three types of experiments. (a) We performed a phase equilibrium study of the system CuO-MgO–GeO<sub>2</sub> in air at 1000°C. (b) We determined the equilibrium oxygen pressures needed for the reduction to Cu<sub>2</sub>O along the (CuMg)O join and for the reduction of  $CuGeO_3$  to  $Cu_2O$  and  $GeO_2$ . This enabled us to calculate the activity-composition relations in the system CuO-MgO and to compute the free energy of formation of CuGeO<sub>3</sub>. (c) By calorimetry at 715°C in a molten oxide solvent, we measured the heat of solution of CuO,  $GeO_2$ , and  $CuGeO_3$  and from that computed the enthalpy of formation of CuGeO<sub>3</sub>.

### 2. Experimental

### 2.1. Phase Equilibrium Studies in Air

The quenching method was used, as described previously (1, 2, 6). Samples of the desired composition were weighed from dried reagent grade CuO, MgO, and GeO<sub>2</sub>, ground

under acetone, and prereacted in platinum crucibles in a muffle furnace at 1000°C. Small samples of this powder were then held in 30Pd-70Ag capsules, or, when coherent pellets could be pressed, fragments thereof were suspended by thin Pt wire in the hot zone of a vertical Kanthal-wound furnace controlled at  $1000 \pm 2^{\circ}C$  for the equilibration runs. After each run, which lasted 48-72 hr, the samples were air-quenched and the phases identified by X-ray diffraction and/or microscopic observation. Landolt and Muan (6) reported the occurrence of quench phases of guggenite in the CuO-CoO system. These formed from solid solutions of rocksalt structure rich in copper. We did not observe any textural differences in any of the guggenite in our samples, and concluded that all the guggenite we saw was at equilibrium at 1000°C. This was borne out by the oxygen pressure measurements in the two phase region (see below). The apparent difference in exsolution kinetics between the CuO-CoO and CuO-MgO systems may be caused by two factors: (a) the more refractory nature of MgO may make all reactions more sluggish than in the CoO-containing system, and (b), the solubility of CuO in the rocksalt phase appears smaller in the MgO-containing system, possibly making the supersaturation during the quench smaller.

The attainment of equilibrium in these runs and in the controlled atmosphere experiments was checked by using several different phase assemblages as starting materials.

### 2.2. Reduction Studies

2.2.1. System CuO-MgO. The technique used was essentially the same as that of Landolt and Muan (6). Nitrogen, purified over copper shot at 250°C, was mixed with compressed air in a conventional gas mixer and the resulting gas mixture was passed upward through the quench furnace. The oxygen pressure for the equilibrium assemblage (CuMg)O, Cu<sub>2</sub>O, O<sub>2</sub> was determined in a succession of runs. Small amounts of Cu<sub>2</sub>O could readily be detected microscopically in grain-mounts of the powdered sample.

2.2.2.  $CuGeO_3$ . Since CuGeO<sub>3</sub> is the only compound in the system CuO-GeO<sub>2</sub>, and

since no compounds are formed in the system  $Cu_2O-GeO_2$  (7), it is possible to measure equilibrium oxygen pressures for the phase assemblage (CuGeO\_3, Cu\_2O, GeO\_2) by methods analogous to those above. The possible small solubility of  $Cu_2O$  in GeO<sub>2</sub> is a complication but the activity of GeO<sub>2</sub> is probably not lowered significantly by this. Speranskaya (7) claims a eutectic in the system  $Cu_2O-GeO_2$  near 950°C, but her diagram is tentative, and we did not observe any liquid or glassy phase in our reduced samples quenched from 1000°C.

Thus, at 1000°C, we determined the oxygen pressure for the equilibrium phase assemblage (CuGeO<sub>3</sub>, Cu<sub>2</sub>O, GeO<sub>2</sub>) by approaching this equilibrium from both sides in a gas-mixer furnace. Because this  $p_{O_2}$  was expected to lie in the range  $10^{-3}$ – $10^{-4}$  atm, using air and nitrogen would require very inconvenient mixing ratios (1:1000). Accordingly, we calibrated our gas mixer with, and used, purified nitrogen passed over Cu at 250°C and a specially prepared and analyzed tank (Matheson) of nitrogen containing 0.1008  $\pm$  0.0010% oxygen. In all other respects, these runs did not differ from the (CuMg)O experiments.

### 2.3. Solution Calorimetry

This technique has been described previously (10, 11). The instrument used was a Calvet-type twin microcalorimeter with a Hastelloy block, recently constructed here. Solution experiments, using molten 2PbO–  $B_2O_3$  at 715 ± 2°C as solvent, were carried out using about 50 mg samples of CuO, GeO<sub>2</sub> (quartz form), and CuGeO<sub>3</sub>. Platinum calibration pieces were used.

# 3. Results and Discussion

# 3.1. Isothermal Section of the System CuO- $MgO-GeO_2$ in Air at $1000^{\circ}C$

This is shown in Fig. 1. Phase relations in the binary system CuO-MgO appear significantly different from those in CuO-CoO. The terminal solubilities in the rocksalt and tenorite phases are much smaller in the magnesium containing system, and the phase labelled guggenite exists at  $0.695 < X_{CuO} < 0.805$ , that is, at more copper-rich compositions and over



FIG. 1. Isothermal section of the system CuO-MgO-GeO<sub>2</sub> in air at  $1000^{\circ}$ C.

a somewhat wider homogeneity range than in the cobalt-containing system. This range includes the formula  $3CuO \cdot MgO$  ( $X_{CuO} =$ 0.75) suggested by Schmahl (5), but not the formula 2CuO·MgO ( $X_{CuO} = 0.667$ ) suggested by Driessens (4) and others. However, it is not very useful to assign rational formulas to compounds exhibiting a wide range of compositions, and the "compound" Cu<sub>2</sub>MgO<sub>3</sub> may well exist at other temperatures. However, the composition CuO·MgO  $(X_{CuO} = 0.5)$ suggested by Trojer (3) clearly falls outside the guggenite range. The composition range  $0.812 < X_{CuO} < 0.86$  reported by Ustyantsev et al. (12) also falls outside our observed guggenite field.

We did not study the structure of the guggenite phase. The lines in our powder X-ray diffraction patterns agreed well with those given by Nitzsche (13). No obvious changes occurred in the diffractograms of samples of different compositions. The growth of single crystals of different compositions and structural work on them would be of interest. In view of the wide homogeneity ranges and the different compositions at which the apparently isostructural guggenite phase is found in the two systems CuO-MgO and CuO-CoO, it appears likely that there is some disorder between Cu and Mg or Cu and Co ions on one or more types of sites in the structure. This idea is supported by the observations of Schmahl and Minzl (5) that guggenite decomposes to rocksalt plus tenorite phases below about 860°C.

The solubility of CuO in the rocksalt phase was found to be 21% at 1000°C, in good agreement with Schmahl (5). This is smaller than the solubility of CuO in CoO (39% (6, 14)) or in NiO (33% (14)).

In the ternary system CuO-MgO-GeO<sub>2</sub>, the olivine phase accepts up to 20% copper substitution. The solubility of CuGeO<sub>3</sub> in the enstatite phase is 36%, and of MgGeO<sub>3</sub> in the CuGeO<sub>3</sub> phase is 7%. The CuGeO<sub>3</sub>-MgGeO<sub>3</sub> join is pseudobinary. The three phase triangles and two phase quadrilaterals are as shown. None of the two phase regions are large enough to permit the determination of tie lines within them.

# 3.2. Activity-Composition Relations in CuO-MgO

The results of reduction experiments are shown in Table I. For the reaction

$$(CuO)_{sol} = \frac{1}{2}Cu_2O + \frac{1}{4}O_2.$$
 (1)  
 $K = (p_{O_2})^{1/4} (a_{Cu_2O})^{1/2} / a_{CuO}.$  (2)

Since  $Cu_2O$  precipitates as an essentially pure phase one may write:

$$(a_{\rm CuO})_{\rm sol} = (p_{\rm O_2}/p_{\rm O_2}^*)^{1/4}$$
(3)

where  $p^*_{O_2}$  is the equilibrium oxygen pressure for the reduction of pure CuO. The activity of MgO may then be calculated by Gibbs-



FIG. 2. Activity-composition relations in the system CuO-MgO at  $1000^{\circ}$ C.

#### TABLE I

EQUILIBRIUM OXYGEN PRESSURES FOR REDUCTION OF CuO IN SYSTEM CuO-MgO-GeO<sub>2</sub> at 1000°C

Initial composition	Phases <sup>a</sup>	P <sub>02</sub> (atm) <sup>b</sup>	a <sub>CuO</sub>
CuO	t+c	1.28 × 10⁻¹	1.000
Cu <sub>0.98</sub> Mg <sub>0.02</sub> O	t+g+c	1.20 × 10 <sup>-1</sup>	0.985
Cu <sub>0.83</sub> Mg <sub>0.07</sub> O	t + 2 + c	1.20 × 10 <sup>1</sup>	0.985
Cu <sub>0.78</sub> Mg <sub>0.22</sub> O	g + c	1.09 × 10 <sup>~1</sup>	0.960
Cu <sub>0,75</sub> Mg <sub>0.25</sub> O	g + c	9.66 × 10 <sup>~2</sup>	0.932
Cu <sub>0.60</sub> Mg <sub>0.40</sub> O	g + r + c	9.17 × 10 <sup>-2</sup>	0.920
Cu <sub>0.20</sub> Mg <sub>0.80</sub> O	r+c	8.78 × 10 <sup>-2</sup>	0.910
Cu <sub>0.15</sub> Mg <sub>0.85</sub> O	r + c	$5.51 \times 10^{-2}$ .	0.810
Cu <sub>0.10</sub> Mg <sub>0.90</sub> O	r + c	2.74 × 10 <sup>-2</sup>	0.680
$CuGeO_3 + GeO_2$	y+q+c	2.22 × 10 <sup>-4</sup>	0.200

<sup>a</sup> t = tenorite solid solution; c = cuprite, Cu<sub>2</sub>O; g = guggenite solid solution; r = rocksalt solid solution; y = CuGeO<sub>3</sub> phase; q = quartz form of GeO<sub>2</sub>.

<sup>b</sup> Every  $p_{02}$  bracketed to approximately  $\pm 1$  % of that value, estimated total error is  $\pm 2$ %.

Duhen integration, and the activity-composition relations are shown in Fig. 2. They agree well with those determined by Schmahl (5) for the rocksalt phase and the two-phase region containing rocksalt and guggenite.

The free energy of formation of guggenite from MgO and CuO may then be calculated over its homogeneity range. This is shown in Fig. 3, together with the curve for hypothetical ideal mixing.

The limiting activity coefficients,  $\gamma^{\circ}$ , of CuO in the rocksalt phase and of MgO in the tenorite phase are 8.00 and 29.2, respectively. We can estimate the free energies of transformation of CuO to the rocksalt structure and of MgO to the tenorite structure from these as:

CuO (tenorite 
$$\rightarrow$$
 rocksalt),  
 $\Delta G^{\circ} \sim RT \ln \gamma^{\circ}_{CuO} = 5.3 \text{ kcal/mole}$  (4)  
and

MgO (rocksalt 
$$\rightarrow$$
 tenorite),  
 $\Delta G^{\circ} \sim RT \ln \gamma^{\circ}_{MgO} = 8.5 \text{ kcal/mole.}$  (5)



FIG. 3. Free energy of formation of guggenite from CuO and MgO at 1000°C.

For the CuO transformation, Schmahl (15) quotes a value of 2.3 kcal/mole, but does not present the data to substantiate it. From his earlier paper (5), however, in which activitycomposition relations in CuO-MgO are presented, and are very similar to those given here, a value of the free energy of transformation very similar to our 5.3 kcal can be calculated. On the basis of calorimetric studies, Navrotsky and Kleppa (16) have suggested that  $\Delta H^0$  for the transformation should be in the range of 4-5 kcal. Since  $\Delta S^0$  for such a solid state transition is probably not more than 1.5 cal/deg mole in magnitude, a  $\Delta G_{1273}^{\circ}$  value of 5.3 kcal is very reasonable. Landolt and Muan (6), on the other hand, estimate  $\Delta G_{1273}^{\circ} = 2.3$  kcal from activity-composition relations in the system CuO-CoO, where the extent of the rocksalt phase is much greater than in CuO-MgO. One must caution, that, in the CuO-MgO system, the very distinct curvature of the  $a_{CuO}$  curve in the rocksalt phase precludes a really accurate determination of the limiting Henry's Law slope, so that the discrepancies between these various studies may be caused by such uncertainties.

The approximate value of  $\Delta G^{\circ}$  for the relation MgO (rocksalt  $\rightarrow$  tenorite) is similar to that for the same hypothetical transformation in CoO (6).

3.3. Enthalpy and Free Energy of Formation of  $CuGeO_3$  from the Oxides

From the reduction studies presented in Table I, we have

$$CuO \rightarrow \frac{1}{2}Cu_2O + \frac{1}{4}O_2, p^*_{O_2} = 0.128 \text{ atm}$$
 (6)  
and

CuGeO<sub>3</sub> = 
$$\frac{1}{2}$$
Cu<sub>2</sub>O + GeO<sub>2</sub> +  $\frac{1}{2}$ O<sub>2</sub>,  
 $p_{O_2} = 2.22 \times 10^{-4}$  atm. (7)

Then for the reaction

CuO + GeO<sub>2</sub> = CuGeO<sub>3</sub>,  

$$K = (p_{O_2}^*/p_{O_2})^{1/4} = 4.90$$
 (8)  
and  $\Delta G^\circ = -RT \ln K = -4.0$  kcal.

The calorimetric data are presented in Table II. The enthalpy of Reaction (8) is equal to the difference in the heats of solution of

$$CuO + GeO_{2} \text{ and of } CuGeO_{3}$$

$$\Delta H_{f}^{\circ} = \Delta H_{sol(CuO)} + \Delta H_{sol(GeO_{2})} - - \Delta H_{sol(CuGeO_{3})}$$

$$= -5.2 \pm 0.4 \text{ kcal/mole.} \qquad (9)$$

Then, the standard entropy of formation of  $CuGeO_3$  from the oxides is:

$$\Delta S^{\circ} = (\Delta H^{\circ} - \Delta G^{\circ})/T = -0.9 \text{ cal/deg mole.}$$
(10)

The fairly substantial negative heat and free energy of formation of  $CuGeO_3$  is consistent with Speranskaya's phase diagram for the system  $CuO-GeO_2$  (7), which shows  $CuGeO_3$ to melt congruently in air some 50° above the temperature at which CuO is reduced to  $Cu_2O$ . The small negative entropy change is characteristic of solid state reactions with no substitutional disorder in reactants or products.

The negative enthalpy of formation of  $CuGeO_3$  is to be contrasted with positive enthalpies of formation for the spinels  $CuFe_2O_4$  (16),  $CuGa_2O_4$  (16),  $CuMn_2O_4$  (16), and  $CuCr_2O_4$  (17). These compounds are "entropy-stabilized" by octahedral-tetrahedral site cation disorder, while the copper silicates, titanates, and orthogermanate do not exist. We have previously attributed these

TABLE II

HEATS OF SOLUTION OF CuO, GeO<sub>2</sub>, and CuGeO<sub>3</sub> in Molten  $2PbO \cdot B_2O_3$  at 713 ± 1°C

(mg) Sample	∆H <sup>m</sup> so1 (kcal/mole)		
CuO (tenorite)			
51.51	8.388		
103.81	8.480		
78.20	8.870		
96.03	8.130		
53.00	8.968		
85.98	8.696		
75.75	8.655		
74.71	8.458		
84.16	8.323		
64.04	8.261		
$av = 8.52 \pm 0.27 (10)^a$			
$GeO_2(quartz)$			
81.87	-4.877		
114.58	-5.031		
83.66	-5.025		
84.35	-5.143		
79.00	-5.027		
69.36	-4.615		
86.30	-4.976		
$av = -4.96 \pm 0.17$ (7)			
CuGeO <sub>3</sub>			
71 12	8 516		
80.53	8 785		
96.02	8.558		
100.80	9.013		
82.17	8.482		
84.94	8.565		
68.62	9.101		
98.82	8.960		
66.12	9.108		
56.86	8.807		
	$av = 8.82 \pm 0.22$ (10)		

<sup>a</sup> Error given is standard deviation, number in parentheses is number of experiments performed. unfavorable energetics of copper spinel formation to the forced change in copper coordination from square planar in tenorite to octahedral and partly tetrahedral in the spinels (16). The negative heat of formation of CuGeO<sub>3</sub>, in which the copper ion retains a coordination similar to that in tenorite, further supports this view.

# 3.4. Discussion: Stability of Ternary Solid Solutions and Structural Transformations

As seen from Fig. 1, none of the two-phase fields (rocksalt plus olivine, guggenite plus olivine, guggenite plus  $CuGeO_3$ , tenorite + CuGeO<sub>3</sub>, olivine + pyroxene, olivine +  $CuGeO_3$ , guggenite +  $CuGeO_3$ , rocksalt + CuGeO<sub>3</sub>) are very extensive. We felt this diagram was thus too complex to apply the method developed by Muan (18) to calculate activity-composition relations along the (CuMg)<sub>2</sub>GeO<sub>4</sub> and (CuMg)GeO<sub>3</sub> joins from the direction of conjugation lines. However, some very approximate calculations can be made as follows. The MgGeO<sub>3</sub>-CuGeO<sub>3</sub> join is pseudobinary, the limits of the enstatite and CuGeO<sub>3</sub> phases occur at  $X_{CuGeO_3} = 0.36$  and 0.93, respectively. If one takes, as an admittedly crude approximation, that the solute follows Henry's Law and the solvent follows Raoult's Law throughout each single phase region, then the activity coefficient of CuGeO<sub>3</sub> in the enstatite phase is  $\gamma_{CuGeO_3} = a_{CuGeO_3}/X_{CuGeO_3} =$ 0.93/0.36 = 2.58, while the activity coefficient of MgGeO<sub>3</sub> in the CuGeO<sub>3</sub> phase is  $\gamma_{MgGeO_3} =$  $a_{MgGeO_3}/X_{MgGeO_3} = 0.64/0.07 = 9.14.$ 

Then for the transformation

CuGeO<sub>3</sub>(CuGeO<sub>3</sub>  $\rightarrow$  enstatite),  $\Delta G^{\circ} = RT \ln \gamma_{CuGeO_3} = +2.4 \text{ kcal/mole}$  (11) and for

MgGeO<sub>3</sub> (enstatite  $\rightarrow$  CuGeO<sub>3</sub>),  $\Delta G^{\circ} = RT \ln \gamma_{MgGeO_3} = +5.6 \text{ kcal/mole.}$  (12)

Both these values are fairly similar to those for the transformations between structures having corresponding coordination in the binary oxides CuO and MgO (see Section 3.2).

The substitution of copper into the olivine phase is quite limited; olivine becomes unstable relative to  $(CuMg)GeO_3$  and guggenite at  $X_{Cu_2GeO_4} = 0.20$ . This suggests that  $Cu_2$ -GeO<sub>4</sub> olivine is unstable with respect to disproportionation into CuGeO<sub>3</sub> and CuO by several (3–5) kcal, which would make its free energy of formation from the oxides zero or slightly positive. A similar conclusion has been reached previously (11) about its heat of formation.

We also performed a few hydrothermal experiments at 730°C and 0.5 kbar water pressure to determine the extent of the  $(CuMg)_2GeO_4$  spinel solid solution, since under those conditions  $Mg_2GeO_4$  has the spinel structure. Preliminary results indicate single phase spinels form for  $X_{Cu_2GeO_4} < \sim 0.2$ . Though no quantitative interpretation can be made, these data suggest that the spinel form of  $Cu_2GeO_4$  would be of fairly comparable stability (or instability) to the olivine.

### Acknowledgments

This work was supported by the National Science Foundation (Grant GP20402 and GH39767). The GeO<sub>2</sub> used was donated by the Sylvania Corporation. D. S. Kenny, L. Ryan, and J. Sharples participated in various aspects of the experimental work.

### References

- 1. A. NAVROTSKY, J. Solid State Chem. 6, 21 (1973).
- 2. A. NAVROTSKY, J. Solid State Chem. 6, 42 (1973).
- 3. F. TROJER, Radex Rundschau, p. 365 (1958).
- 4. F. C. M. DRIESSENS, G. D. RIECK, AND H. N. COENEN, J. Inorg. Nucl. Chem. 30, 747 (1968).
- 5. N. G. SCHMAHL AND E. MINZL, Z. Phys. Chem. (N.F.) 41, 66 (1964).
- C. LANDALT AND A. MUAN, J. Inorg. Nucl. Chem. 31, 1319 (1969).
- 7. E. I. SPERANSKAYA, *Izv. Akad. Nauk S.S.S.R.*, *Neorg. Mater.* **3**, 1458 (1967).
- H. VÖLLENKLE, A. WITTMAN, AND H. NOWOTNY, Monats. Chem. 98, 1362 (1967).
- 9. A. M. M. GADALLA, W. F. FORD, AND J. WHITE, *Trans. Brit. Ceram. Soc.* **62**, 62 (1963).
- 10. O. J. KLEPPA, Proc. Brit. Ceram. Soc., 8, 31 (1967).
- 11. A. NAVROTSKY, J. Inorg. Nucl. Chem. 33, 4035 (1971).
- 12. V. M. USTYANTSEV AND A. F. BESSONOV, Zh. Prikl. Khim. 41, 1443 (1968).

- 13. H. G. NITZSCHE, Radex Rundschau, p. 383 (1963).
- 14. N. G. SCHMAHL AND G. F. EIKERLING, Z. Phys. Chem. (N.F.) 62, 268 (1968).
- 15. N. G. SCHMAHL AND E. MINZL, Z. Phys. Chem. (N.F.) 47, 142 (1965).
- 16. A. NAVROTSKY AND O. J. KLEPPA, J. Inorg. Nucl. Chem. 30, 479 (1968).
- 17. F. MÜLLER AND O. J. KLEPPA, J. Inorg. Nucl. Chem. 35, 2673 (1973).
- 18. A. MUAN, Proc. Brit. Ceram. Soc. 8, 103 (1967).